

Stable Mott Polaron State Limits the Charge Density in Lead Halide Perovskites

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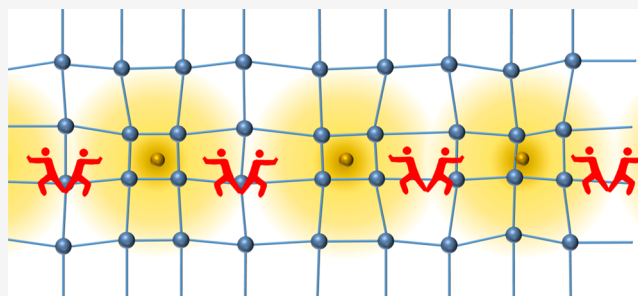


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Supporting Information

ABSTRACT: Large polarons are known to form in lead halide perovskites (LHPs). Photoinduced isolated polarons at low densities have been well-researched, but many-body interactions at elevated polaron densities, exceeding the Mott criterion (i.e., Mott polaron density), have remained elusive. Here, employing ultrafast terahertz spectroscopy, we identify a stable Mott polaron state in LHPs at which the polaron wavefunctions start to overlap. The Mott polaron density is determined to be $\sim 10^{18} \text{ cm}^{-3}$, in good agreement with theoretical calculations based on the Feynman polaron model. The electronic phase transition across the Mott density is found to be universal in LHPs and independent of the constituent ions. Exceeding the Mott polaron density, excess photoinduced charge carriers annihilate quickly within tens to hundreds of picoseconds, before reaching the stable and long-lived Mott state. These results have considerable implications for LHP-based devices and for understanding exotic phenomena reported in LHPs.



Low-temperature solution-processed lead halide perovskites (LHPs) have shown superior performance in various electro-optic applications, such as light-emitting diodes (LEDs)^{1,2} and photovoltaic cells.^{3,4} These devices rely on the interconversion of photons and mobile charge carriers. Due to the ionic and soft nature of the perovskite structure, the crystal lattice is readily polarized and distorted around an injected charge carrier. The charge carrier becomes thus dressed by a local lattice deformation, forming a polaron quasiparticle. Because of the weak-to-intermediate electron–phonon interaction in LHPs, large polarons develop, with the lattice distortion spanning over multiple lattice constants, i.e., a few nanometers. Large polaron formation has been proposed in LHPs to underpin their fascinating optoelectronic properties,⁵ including long carrier lifetimes,⁶ good charge mobilities,⁷ and high defect tolerance.⁸ The polarized crystal structure could act as a protective “shield” to effectively screen the Coulombic interactions between the charge carriers and ionized defects, phonons, and among themselves. So far, the spectroscopic and transport studies of polarons in LHPs have mainly focused on the low polaron density regime ($\sim 10^{16} \text{ cm}^{-3}$),^{9–11} relevant for, for instance, solar cell applications, where large polarons are, on average, well-separated.

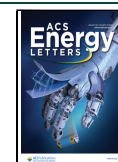
LHPs have also shown great potential in high-excitation optoelectronic applications, including LEDs,¹ lasing,¹² and concentrator photovoltaics,¹³ where the photoinduced carrier

density commonly exceeds 10^{18} cm^{-3} . In this high-density regime, the distorted lattice pattern from different polarons may start to interfere as polarons approach each other. The resultant polaron–polaron interaction significantly impacts the optoelectronic properties, including charge transport and carrier lifetime. For example, strong Auger recombination is typical for LHPs, leading to an efficiency roll-off in, e.g., LEDs.^{14,15} Extremely long-lived hot carriers with lifetime $>100 \text{ ps}$ have been reported in LHPs¹⁶ at carrier densities exceeding 10^{18} cm^{-3} . Frost et al. attributed this delayed hot-carrier cooling at high fluence to a Mott transition, defined as the carrier density where polaron wavefunctions overlap.¹⁷ This critical density is also referred to as the Mott polaron density,^{17,18} being analogous to the metal–nonmetal transition that is observed in doped semiconductors.¹⁹ However, the specific Mott transition and the transport properties in this high-excitation regime have remained largely unexplored

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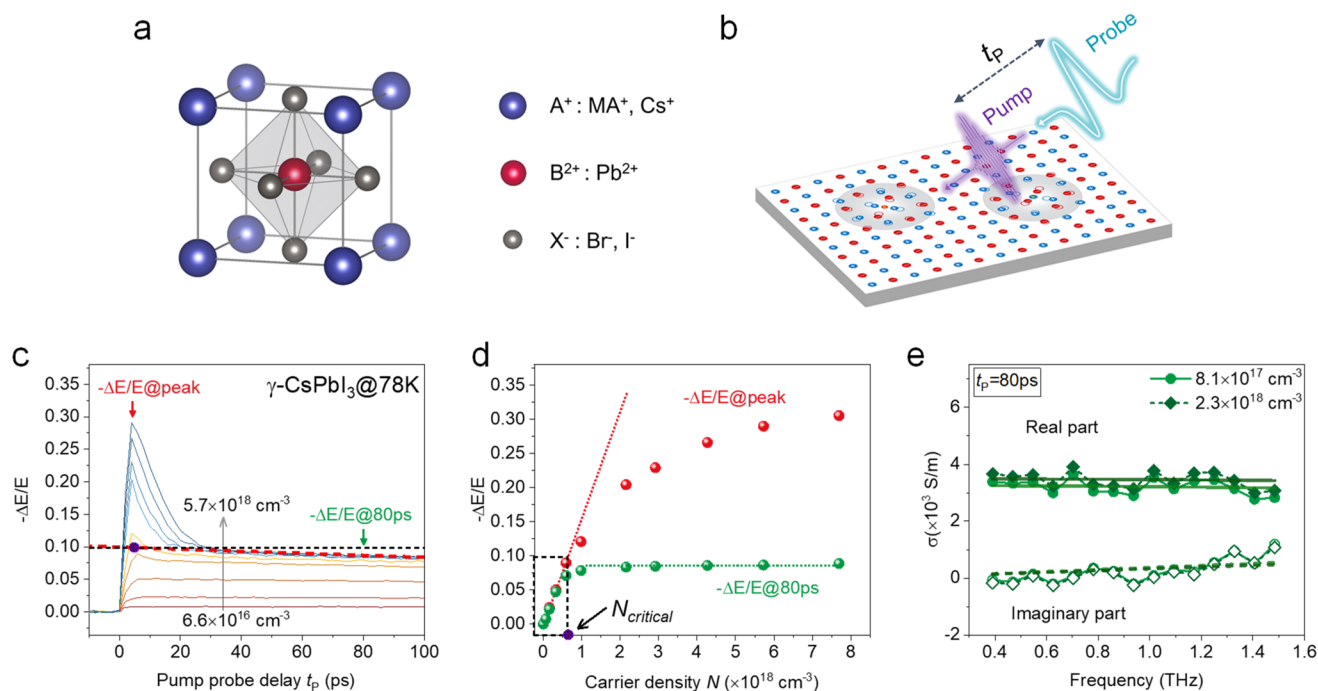


Figure 1. Carrier density saturation in black γ -CsPbI₃. (a) Atomic unit cell of lead halide perovskites; the three different samples used for this study are γ -CsPbI₃, MAPbI₃, and CsPbBr₃. (b) Schematic illustration of the optical pump–THz probe (OPTP) experiment. (c) Time-dependent $-\Delta E/E$ signals (reflecting carrier densities) at different fluences, in black γ -CsPbI₃ at 78 K, with a pump photon energy of 3.10 eV. The red dashed line is a linear extrapolation of the slow decay between 40 ps and 100 ps at the highest fluence to zero delay. The black dashed line denotes the corresponding signal at the peak $-\Delta E/E$ (the purple dot). (d) Intensity of $-\Delta E/E$ at the peak and at $t_p = 80$ ps in (c) as a function of the initially photoinjected carrier density. The red dotted line is the linear fit to the peak $-\Delta E/E$ values in the low carrier density regime. The green dashed line is the constant fit to $-\Delta E/E$ at $t_p = 80$ ps in the high carrier density regime. N_{critical} is defined as the carrier density at which the peak $-\Delta E/E$ value is the same as the cross point (the purple dot) in (c). (e) Frequency-resolved complex photoconductivity spectra consisting of real and imaginary components measured at $t_p = 80$ ps with initial carrier densities of $8.1 \times 10^{17} \text{ cm}^{-3}$ and $2.3 \times 10^{18} \text{ cm}^{-3}$. The solid and dashed lines represent the Drude fits.

experimentally for LHPs, despite their importance and relevance for many applications.

In this work, employing ultrafast terahertz (THz) spectroscopy, we report the observation of an exotic, stable, and long-lived Mott polaron state in LHPs, at which the polaron wavefunctions start to overlap. We quantify the Mott polaron density N_{Mott} to be $\sim 10^{18} \text{ cm}^{-3}$, in line with theoretical calculations from a numerical solution to the Feynman polaron model. The electronic phase transition across the Mott density is found to be universal in LHPs and independent of the detailed chemical composition, representing an intrinsic property of LHPs. With increasing excitation density, the excess photoinjected charge carriers beyond N_{Mott} annihilate quickly, within tens to hundreds of picoseconds depending on the temperature, to eventually relax to the stable Mott state. These results shed light on the intrinsic polaron many-body effects in LHPs, which impact light-concentrated optoelectronic devices.

LHPs have the general chemical formula of ABX₃, in which A is an organic or inorganic monovalent cation (e.g., methylammonium MA⁺, formamidinium FA⁺, or Cs⁺), B is a bivalent cation (here, lead), and X is a monovalent halide anion (such as Cl⁻, Br⁻, I⁻). As shown in Figure 1a, the crystal structure of LHPs consists of corner-shared octahedra [BX₆]⁴⁻ with the A cations sitting in the inter-octahedral voids. In this work, three LHPs (i.e., MAPbI₃, γ -CsPbI₃, and CsPbBr₃) are prepared as model systems and spin-coated on the fused silica substrates for THz measurements (see Methods for sample

preparation). The UV–vis absorption spectra are measured and shown in Figure S1, and the inferred optical properties (e.g., bandgap) are consistent with previous reports. During the THz measurements, the samples are kept under a high vacuum in a cryostat with a background pressure of $\sim 2 \times 10^{-4}$ mbar. We observe no sample degradation during the THz measurements for all samples. More particularly, the optically active black orthorhombic phase γ -CsPbI₃ remains intact.

To investigate the polaron dynamics in LHPs, we use contact-free optical pump–THz probe (OPTP) spectroscopy and THz time-domain spectroscopy (THz-TDS). A detailed description of both measurements is included in the Methods. Briefly, as shown in Figure 1b, in the OPTP measurements, the perovskite sample is first excited by an ultrashort optical (“pump”) pulse with 50 fs pulse duration to photoinject charge carriers into the conduction band (i.e., electrons) and valence band (i.e., holes). After a controlled delay time t_p , a single-cycle THz pulse as the “probe” with ~ 1 ps duration propagates collinearly through the excited sample and interacts with the photoinjected charge carriers. This interaction causes an attenuation of the THz field. The attenuation provides a direct measure of the conductivity σ of the sample, which in turn is defined by the product of the elementary charge e , the carrier density N , and the carrier mobility μ : $\sigma = eN_e\mu_e + eN_h\mu_h$ (with $N_e = N_h$, where the subscripts “e” and “h” stand for electrons and holes, respectively). As such, THz spectroscopy provides direct access to the charge density following optical excitations, with high time resolution (~ 200 fs).^{20,21} The

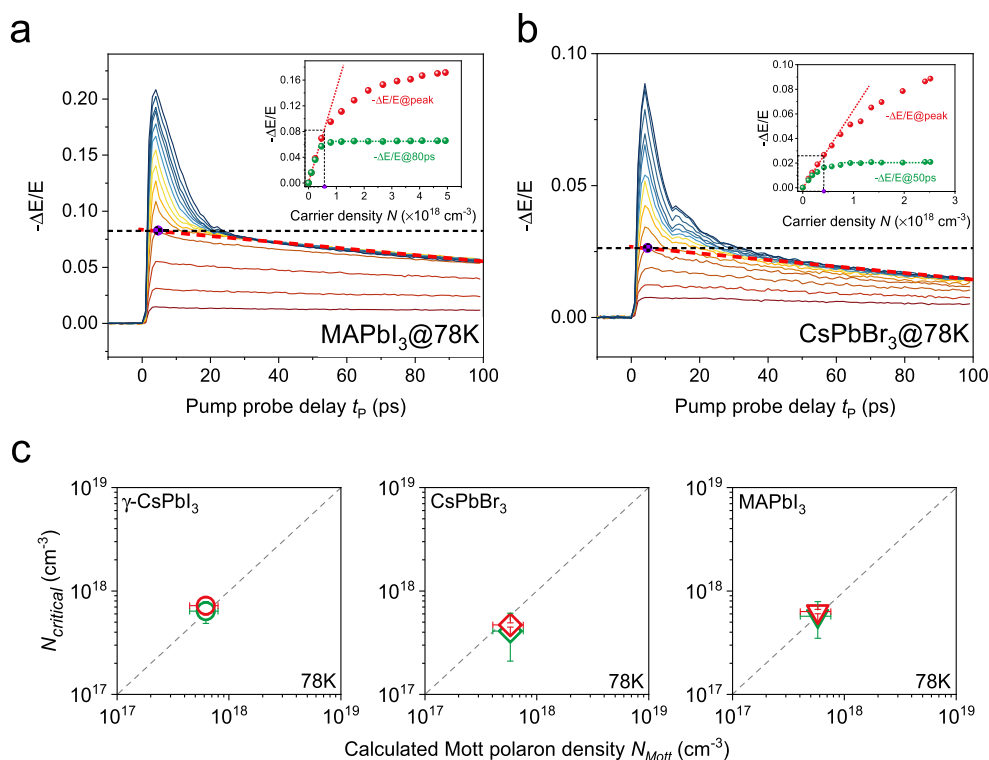


Figure 2. THz measurements in MAPbI₃ and CsPbBr₃. (a, b) Fluence-dependent OPTP dynamics in MAPbI₃ and CsPbBr₃ at 78 K, with an excitation photon energy of 3.10 eV. Insets: the intensity of $-\Delta E/E$ at the peak and at later time delays ($t_p = 80$ ps for MAPbI₃ and $t_p = 50$ ps for CsPbBr₃) as a function of photoinjected carrier density. (c) Comparison between the experimentally extracted critical density N_{critical} and the calculated Mott polaron density based on the Feynman polaron model for γ -CsPbI₃, MAPbI₃, and CsPbBr₃. The green symbols represent N_{critical} obtained from the OPTP measurements, while the red symbols are N_{critical} extracted from the THz-TDS measurements. The dashed lines indicate $N_{\text{critical}} = N_{\text{Mott}}$.

transmitted THz electric field following the excitation pulse $E_{\text{exc}}(t)$ is detected coherently by another 800 nm sampling pulse (pulse duration 50 fs) in a ZnTe crystal via the electro-optic effect. The transmitted field in the absence of the excitation pulse $E_0(t)$ is also determined. It can be demonstrated that the quasi-instantaneous sample conductivity at time t_p can be approximated by $\sigma(t_p) \propto -\Delta E(t_p)/E_0$, with $\Delta E(t_p) = E_{\text{exc}}(t_p) - E_0$.²² In addition, by mapping out the change of the full THz waveform at a fixed pump-sampling delay time t_p and then performing the Fourier transformation, the complex frequency-resolved photoconductivity spectrum $\sigma(\omega)$ at t_p can be obtained.

In Figure 1c, we show the fluence-dependent OPTP dynamics of black γ -CsPbI₃ at 78 K with varying photoinjected carrier density N over nearly 2 orders of magnitude (from $6.6 \times 10^{16} \text{ cm}^{-3}$ to $5.7 \times 10^{18} \text{ cm}^{-3}$). N is inferred from the incident photon density N_{photon} , the sample absorption A , and the photon-to-charge quantum yield Φ : $N = \Phi N_{\text{photon}} A$. Φ is around 30% for black γ -CsPbI₃.¹⁰ An independent determination of the quantum yield from the plasma frequency obtained from the THz conductivity spectra confirms this value (see Supporting Information). Due to their similar effective masses,²³ both electrons and holes are included here, as they contribute almost equally to the measured photoconductivity. With increasing the carrier density N , distinct features appear in the OPTP dynamics: in the low excitation regime ($N < \sim 7 \times 10^{17} \text{ cm}^{-3}$), the amplitude of the photoconductivity increases linearly with excitation density and shows little decay within our time window of ~ 1 ns (see the inferred lifetime at low fluences in Figure S3 in the

Supporting Information). The observed low charge carrier recombination rate is in line with our previous report on the carrier dynamics of γ -CsPbI₃,¹⁰ and is also consistent with the picture of large polaron formation, which reportedly screens the charge carriers from defects and other charge carriers.^{6,10,20,24} When the polaron density is low, the overlap between their wavefunctions is small, thus limiting the bimolecular recombination rate.

In the high excitation regime (with $N > \sim 7 \times 10^{17} \text{ cm}^{-3}$), a rapid decay appears with a time constant spanning from several to tens of ps, then followed by a long-lived signal again as in the low excitation regime (see Figure S3 in the Supporting Information for fluence-dependent OPTP dynamics at long time scales). The peak photoconductivity increases continuously with photoinjected carrier density, yet the signals quickly decay to the same level, independent of the initial excitation density. This unique carrier-density-dependent photoconductivity evolution is summarized in Figure 1d by plotting the signal amplitude at two representative time cuts: at the photoconductivity peak, and at 80 ps after photoexcitation. This figure reveals the existence of a critical charge carrier density N_{critical} : below N_{critical} , the photoconductivity increases linearly with N , and it is long-lived; exceeding N_{critical} , the peak photoconductivity is still increasing with N , yet the conductivity at later times (e.g., $t_p = 80$ ps) reaches a plateau. To accurately estimate the critical density N_{critical} , we extrapolate the long-lived signal at the highest fluence (from 40 ps to 100 ps here for γ -CsPbI₃) back to the peak $-\Delta E/E$ value, as shown in Figure 1c, and then project this value to the corresponding carrier density N , i.e., critical density N_{critical} by

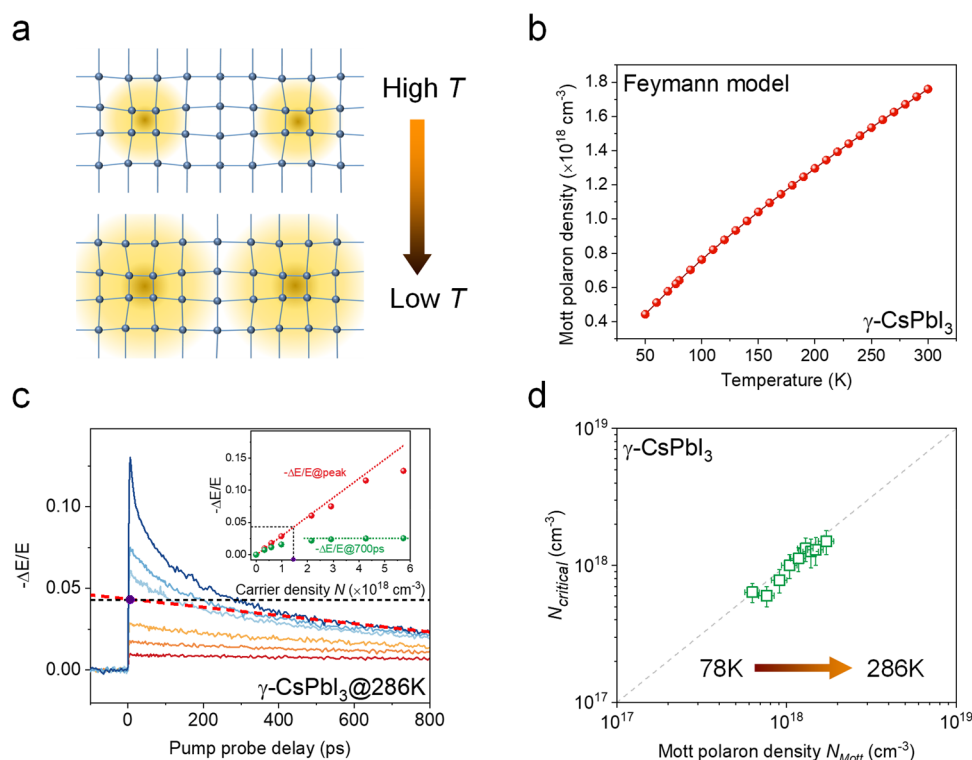


Figure 3. Temperature-dependent critical density in LHPs. (a) Schematic illustration of polaron size evolution upon lowering the temperature. (b) Calculated Mott polaron density as a function of temperature based on the finite temperature Feynman polaron model for black $\gamma\text{-CsPbI}_3$. (c) Fluence-dependent OPTP dynamics at 286 K in black $\gamma\text{-CsPbI}_3$ with an excitation photon energy of 3.10 eV. Insets: the intensity of $-\Delta E/E$ at the peak and at $t_p = 700 \text{ ps}$ as a function of photoinjected carrier density. The dashed red line shows the extrapolation back to zero pump–probe delay from data between 500 ps and 800 ps at high excitation density, as in Figure 1c. (d) Comparison between the experimentally (OPTP) extracted critical density N_{critical} and the calculated Mott polaron density at different temperatures for $\gamma\text{-CsPbI}_3$. The dashed line indicates where N_{critical} and N_{Mott} are the same.

using the linear relation between the peak value of $-\Delta E/E$ and N at low fluences. For black $\gamma\text{-CsPbI}_3$ investigated here, we find $N_{\text{critical}} = (6.4 \pm 1.5) \times 10^{17} \text{ cm}^{-3}$.

To verify if the long-time photoconductivity for all excitation fluences reaches the same state (with the same density and mobility of charge carriers), we measured full photoconductivity spectra at $t_p = 80 \text{ ps}$. Figure 1e shows two typical complex photoconductivity spectra under two different excitation densities (with $N = 8.1 \times 10^{17} \text{ cm}^{-3}$ and $2.3 \times 10^{18} \text{ cm}^{-3}$). The complex spectra share nearly the same dispersion and intensity (see more data and discussion below). This result indicates that, independent of initial excitation density, for $t_p > 30 \text{ ps}$, the charge carriers reach the same state with the same charge mobility and carrier density, independent of the initial excitation density. We tentatively attribute the fast carrier loss during the first 30 ps in the high excitation regime to Auger recombination, as detailed in the Supporting Information, but cannot exclude other contributions to the decay of the signal.

At the same time, THz-TDS provides an independent way to estimate the critical density, that is, to fit the photoconductivity spectrum by theoretical models (e.g., Drude model or Drude–Smith model) and extract the carrier density (see the spectra and fitting for all three samples in the Supporting Information). For black $\gamma\text{-CsPbI}_3$, we fit the photoconductivity spectrum (measured at $t_p = 80 \text{ ps}$ at an excitation fluence of $9.35 \times 10^{17} \text{ cm}^{-3}$) by the Drude model. After correcting for the small density decay occurring up to 80 ps, we find $N_{\text{critical}} = (7.2 \pm 1.4) \times 10^{17} \text{ cm}^{-3}$, in good

agreement with the value obtained from the OPTP result. Importantly, we find the critical density is independent of the excitation photon energy (see a comparison of 400 nm and 750 nm excitation in the Supporting Information). This result further confirms the intrinsic origin of the observed critical density.

To test whether the observed carrier density saturation is unique to the black $\gamma\text{-CsPbI}_3$, we conducted measurements on the other two typical LHPs, i.e., MAPbI_3 and CsPbBr_3 , with either the A-site cation or the X-site anion changed compared to $\gamma\text{-CsPbI}_3$. Although some details about OPTP dynamics vary for different perovskites (e.g., the carrier lifetime and signal amplitude), the same carrier density saturation phenomenon is observed in MAPbI_3 and CsPbBr_3 when the carrier density exceeds a threshold N_{critical} , as shown in Figure 2a,b. Based on this observation, we can conclude that this effect seems universal in LHPs, including organic–inorganic hybrids and all-inorganic LHPs, independent of the ionic composition. By the same approach utilized in black $\gamma\text{-CsPbI}_3$ to extract the critical density from OPTP measurement, the insets of Figure 2a,b show that the obtained N_{critical} for MAPbI_3 and CsPbBr_3 are $(5.7 \pm 2.2) \times 10^{17} \text{ cm}^{-3}$ and $(4.1 \pm 2.0) \times 10^{17} \text{ cm}^{-3}$, respectively. By THz-TDS, we extract the N_{critical} for MAPbI_3 and CsPbBr_3 as $(5.8 \pm 0.3) \times 10^{17} \text{ cm}^{-3}$ and $(4.7 \pm 0.2) \times 10^{17} \text{ cm}^{-3}$, respectively, taking the density decay into consideration as well (see Supporting Information).

Converting the critical density to the average volume per charge carrier and further the distance d between neighboring carriers, we estimate d to be around 10 nm. This value is

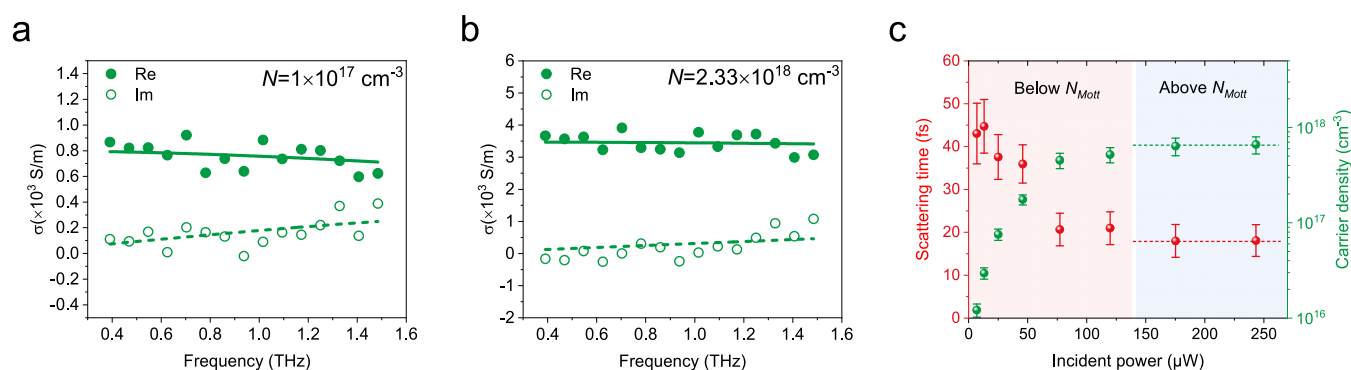


Figure 4. Fluence-dependent complex photoconductivity for black γ -CsPbI₃ measured at $t_p = 80$ ps at a temperature of 78 K. (a, b) Frequency-resolved photoconductivity spectra (filled and empty circles) and Drude model fits (solid and dashed lines) at initial photogenerated carrier densities of $1 \times 10^{17} \text{ cm}^{-3}$ and $2.33 \times 10^{18} \text{ cm}^{-3}$. (c) Extracted scattering times and carrier densities from the Drude fits at different incident fluences.

similar to the polaron diameter in LHPs.^{17,25} This simple estimate indicates that N_{critical} can be interpreted as the Mott polaron density N_{Mott} in LHPs in which neighboring polarons overlap, and it sets the upper limit of the charge carrier density that LHPs can support in thermal equilibrium. To examine this hypothesis, in Figure 2c we correlate the experimentally extracted critical densities N_{critical} (from both OPTP and THz-TDS) to the calculated N_{Mott} for all three perovskites at 78 K. The Mott polaron density is calculated based on a finite temperature numeric solution to the Feynman polaron model.²⁶ The polaron radius is defined following Schultz as the width of the Gaussian polaron wavefunction ansatz in the Feynman model.²⁷ Assuming, for simplicity, that the polaron occupies a cube with sides twice the polaron radius, the Mott polaron density is thus inferred. The parameters used to calculate N_{Mott} for the three perovskites are shown in Table S3 in the Supporting Information. As can be seen in Figure 2c, a direct correlation between the extracted critical density N_{critical} and the Mott polaron density N_{Mott} is evident for all three LHPs.

To further test the one-to-one correlation between N_{critical} and N_{Mott} we perform temperature (T)-dependent photoconductivity studies. Based on the Feynman theory, the polaron radius, and thereby N_{Mott} is strongly temperature-dependent. In the finite temperature formulation of Feynman's polaron model, the maximum extent of the polaron is at zero temperature. At increasing temperature, the finite phonon population (and associated polarization fields) further localizes the charge carrier. This relationship, and the accuracy of the Feynman method, was confirmed in a recent quantum Monte Carlo study.²⁸ In other words, the polaron size increases as T decreases,¹⁷ as shown schematically in Figure 3a. Correspondingly, the model predicts that the Mott polaron density will increase with T . Figure 3b shows the calculated N_{Mott} at different T for black γ -CsPbI₃. The calculated Mott polaron density increases substantially, by a factor of ~ 4 , from 50 K to 300 K. To check the effect of T on the OPTP dynamics and critical density, Figure 3c shows the fluence-dependent OPTP measurement in black γ -CsPbI₃ at 286 K as an example (see more T -dependent measurements in the Supporting Information), and the inset displays the extraction of the critical density. With increasing initial excitation density, the signal saturation takes place in hundreds of ps at 286 K, instead of tens of ps at 78 K. The extracted critical density here is $(1.47 \pm 0.3) \times 10^{18} \text{ cm}^{-3}$, which is very close to the calculated Mott

density of $1.67 \times 10^{18} \text{ cm}^{-3}$ at 286 K. In Figure 3d, we further show the inferred N_{critical} at T varying from 78 K to 286 K, which correlates well with N_{Mott} obtained from calculations. Note that there is no phase transition²⁹ for black γ -CsPbI₃ below room T , and we observe nearly no change in the photon-to-carrier conversion efficiency within this T range (see Supporting Information). This T -dependent result further supports the existence of a Mott polaron transition in LHPs. This delayed signal saturation at high T indicates a slower stabilization at the Mott polaron density, which may be due to increased thermal disorder (anharmonic phonon scattering) at elevated temperatures. The higher Mott density impedes thermal dissipation from hot polarons.¹⁷

After establishing the Mott polaron state and quantifying its density, we characterize the charge transport properties below and above it. In Figure 4a,b, we show two typical photoconductivity spectra of the black γ -CsPbI₃ at 78 K following two excitation fluences, below and above the Mott density: $1 \times 10^{17} \text{ cm}^{-3}$ and $2.33 \times 10^{18} \text{ cm}^{-3}$. Both measurements are conducted at 80 ps following optical excitations to ensure that Mott polaron states are established following exceeding-Mott-density excitations. As we can see, both photoconductivity spectra share the same feature, with a positive, decreasing real part and a positive, increasing imaginary part with frequency. This is a clear signature of delocalized band transport and can be well fitted by the Drude model. The solid and dashed lines in Figure 4a,b show the real and imaginary conductivity, respectively, and their description using the Drude model. The Drude model has two free parameters: carrier density N and carrier scattering time τ . Figure 4c summarizes the excitation-density-dependent N and τ . The obtained carrier density grows rapidly and then saturates at $\sim 6.4 \times 10^{17} \text{ cm}^{-3}$, consistent with the inferred Mott density above in γ -CsPbI₃ at 78 K. Furthermore, the extracted scattering time shows a strong fluence-dependent transition, from an almost fluence-independent high scattering value of over 40 fs at low polaron density (for absorbed photon density below $\sim 40\%$ of Mott density) to less than 20 fs above the Mott density. This result indicates that even at the Mott polaron density, delocalized band transport still prevails in LHPs, but that some additional scattering pathway has been activated. Most likely, this is carrier-carrier scattering now that the polaron wave functions are overlapping. The enhanced carrier scattering interaction reduces the charge mobility ($\mu = e\tau/m^*$, with m^* as the effective mass) from $\sim 570 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ below to $\sim 350 \text{ cm}^2 \text{ V}^{-1}$

s^{-1} above the Mott density, despite a maximum $\sim 25\%$ reduction in effective mass above the Mott density (see Supporting Information).

The observation of the Mott polaron state sets the upper limit on the available large polaron population (i.e., N_{Mott}) that LHPs can host. Theoretically, Emin predicted an emerging repulsive interaction acting as an energy barrier between the oppositely charged large polarons when the polaron separation is small enough.³⁰ Such repulsive polaronic interactions are balanced by the electron–hole Coulomb attraction, determining the spatial energy landscape of polaron states. These balanced interactions underlie the formation of the stable Mott polaron state in LHPs. We note that the “ferroelectric” nature of the polarons proposed by Zhu et al. may further impact the electrostatic stabilization of the Mott polaron state.³¹ Besides the low trapping rate (so-called “defect tolerance” effect) reported in LHPs linked to the dielectric screening that also drives large polaron formation, our discussion here can also explain the weak bimolecular recombination rate (i.e., free electron–hole recombination) and thus long carrier lifetime in LHPs in solar cell applications.

At high excitation densities ($N > 10^{18} \text{ cm}^{-3}$), unexpectedly strong Auger recombination in LHPs is often deduced from spectroscopic studies, including THz spectroscopy^{10,32} and transient absorption spectroscopy.^{33,34} The inferred Auger coefficient is almost 2 orders of magnitude higher in LHPs than in conventional semiconductors with similar bandgaps.¹⁴ This strong Auger recombination leads to the efficiency roll-off in LEDs¹⁵ and concentrator photovoltaics¹³ when the carrier density is above $\sim 10^{18} \text{ cm}^{-3}$. In addition, an exceptionally accelerated dynamics decay is observed at lower temperatures.^{32,35} Yet, the underlying mechanism for the strong Auger recombination in LHPs is still under debate. These phenomena can be rationalized by the formation of the stable Mott polaron state observed in this work. For densities exceeding N_{Mott} the enforced overlap of polarons weakens the protection of the induced charges.³⁶ This results in an enhanced electron–hole interaction and, therefore, a faster population annihilation rate.¹¹ At lower temperatures, the population reduces to the Mott polaron density faster, within tens of ps, in line with the enhanced Auger recombination at low T .

The observed Mott density may also be closely related to the carrier densities at which long hot carrier lifetimes were reported in LHPs. At a carrier density of $6 \times 10^{18} \text{ cm}^{-3}$, Yang et al. reported hot carrier lifetimes up to 100 ps in MAPbI₃ and FAPbI₃.¹⁶ Frost et al. proposed that this slowed hot carrier cooling at high fluence originates from the polaron overlap and shared phonon subpopulation.¹⁷ Our results corroborate this proposal. Due to the destabilized polaron protection above the Mott polaron density, the hot carrier–longitudinal optical (LO) phonon interaction increases with fluence. However, polaron overlap at the stable Mott state impedes the efficient phonon diffusion away from the hot carriers, which, in turn, reheats the relaxed carrier population. As a consequence, the hot carriers in LHPs can be maintained for a long time.

In summary, we observe the formation of a stable Mott polaron density in lead halide perovskites in the high-excitation regime ($>10^{18} \text{ cm}^{-3}$). This phenomenon is universal and independent of the constituents' ionic nature. Quantitative agreement is found for the Mott polaron density between the experiments and the calculations from the temperature-dependent Feynman variational polaron theory. Above the

Mott polaron density, the photoinjected excess carriers annihilate within tens to hundreds of ps depending on the temperature. Our results are crucial for understanding the intrinsic optoelectronic properties of LHPs and shed light on the performance of light-concentrated optoelectronic devices.

METHODS

Synthesis of Lead Halide Perovskites. All reagents were used as received without any further purification. Methylammonium iodide >99.99% (Greatcell Solar Materials); lead(II) iodide ultra-dry 99.999% (metals basis) (Thermo Scientific); *N,N*-dimethylformamide (DMF) 99.8% extra dry over AcroSeal molecular sieves (Acros Organics); cesium bromide 99.9% metals basis (Thermo Scientific); lead(II) bromide $\geq 98\%$ (Sigma-Aldrich); dimethyl sulfoxide (DMSO) 99.7+% extra dry over AcroSeal molecular sieves (Acros Organics); chlorobenzene 99+% for spectroscopy (Acros Organics); γ -butyrolactone (GBL) ReagentPlus $\geq 99\%$ (Sigma-Aldrich); and cesium iodide 99.9% trace metals basis (Sigma-Aldrich).

The substrates for the film deposition consisted of fused silica substrates (1 cm \times 1 cm) which were washed with isopropanol, dried, and left in an ozone reactor chamber for 30 min.

Thin films of CH₃NH₃PbI₃ were prepared by a modified protocol according to a previous report.¹ CH₃NH₃I (0.0795 g) and PbI₂ (0.2301 g) were mixed in a 1:1 stoichiometric ratio in anhydrous DMF (1 mL) to give a 0.5 M solution, which after stirring at room temperature for at least 20 min produced a clear CH₃NH₃PbI₃ solution. The solution was filtered with a 0.20 μm pores hydrophilic PTFE filter. The perovskite film deposition was performed under an inert atmosphere inside a glovebox. The CH₃NH₃PbI₃ solution (50 μL) was deposited on the fused silica glass and then spin-coated at 5000 rpm for 30 s. The solvent was evaporated at 100 $^{\circ}\text{C}$ for 10 min (still in the glovebox), resulting in a clear brown film. The more diluted CH₃NH₃PbI₃ film was obtained by diluting 15 μL of the former CH₃NH₃PbI₃ solution in 45 μL of anhydrous DMF, keeping the rest of the protocol the same. The film was stored in the dark under an inert atmosphere until further measurements.

CsPbBr₃ thin films were prepared by spin-coating a 0.5 M DMSO solution of CsBr and PbBr₂ (1:1 stoichiometric ratio). The solution was prepared by mixing CsBr (0.1064 g) and PbBr₂ (0.1835 g) in 1–1.5 mL of dry DMSO at around 70 $^{\circ}\text{C}$. The solution was then filtered with a 0.20 μL pore PTFE hydrophilic filter. For obtaining the thin film, 70 μL of the perovskite solution was spin-coated on the fused silica glass, in a glovebox. Subsequent spin-coating at 1000 rpm for 10 s, followed by 60 s at 3000 rpm, with the addition of 125 μL of chlorobenzene 35 s before the end of the program, resulted in the desired thin film after solvent evaporation (for 10 min on a hot plate at 60 $^{\circ}\text{C}$). The diluted CsPbBr₃ thin film was prepared by diluting the former perovskite solution by 25% in volume, while retaining all other steps. Both of the films were annealed outside the glovebox for 30 min at 150 $^{\circ}\text{C}$. The films were stored in the dark under an inert atmosphere until further measurements.

CsPbI₃ thin films were prepared from a 1:1 stoichiometric solution of CsI and PbI₂. The two salts were mixed in a 4:1 DMSO:GBL solution (1 M) at 60 $^{\circ}\text{C}$ overnight and then filtered with a 0.20 μL pore PTFE hydrophilic filter. 70 μL of the clear solution was spin-coated on the substrate using the

following program: 300 rpm for 30 s, 1000 rpm for 20 s, and finally 4000 rpm for 60 s. The coated substrate was heated first at 160 °C for 1 h and then at 320 °C for 15 min, until the substrate changed color to the desired black phase. The whole procedure was carried out under an inert atmosphere (N₂) in a glovebox. The samples were stored in the dark under an inert atmosphere until for further measurements.

Optical Pump–THz Probe (OPTP) Spectroscopy. The OPTP setup is driven by a commercial, regenerative amplified, mode-locked Ti:sapphire femtosecond laser with 1 kHz repetition rate. In the OPTP measurement, as shown in Figure 1b, the perovskite sample is first pumped by a 400 nm (~3.10 eV) optical pulse with a 50 fs pulse duration. The pump pulse is generated from an 800 nm (~1.55 eV) pulse by second-harmonic generation in a beta barium borate (BBO) crystal. After a pump–probe delay time t_p , the THz pulse propagates collinearly with the pump pulse and transmits through the sample. The single-cycle THz pulse is generated from an 800 nm pulse by optical rectification in a 1-mm-thick ZnTe crystal. The transmitted THz electric field waveform $E(t)$ is detected coherently in a second ZnTe crystal by another 800 nm sampling pulse via the electro-optic effect. The photoconductivity dynamics $\sigma(t_p)$ of the sample is thus obtained by reading out the pump-induced relative change of THz peak electric field $-\Delta E/E_0(t_p)$ as a function of t_p , based on the thin-film approximation:³⁷

$$\sigma(t_p) = \frac{n_1 + n_2}{Z_0 l} \left[-\frac{\Delta E}{E_0}(t_p) \right]$$

Here, $\Delta E = E_{\text{exc}} - E_0$, in which E_0 and E_{exc} are the transmitted THz electric fields before and after the pump. n_1 and n_2 are the refractive indices in front of and behind the sample. Z_0 is the free space impedance. l is the thin-film thickness (~200 nm for all films in this study).

THz Time-Domain Spectroscopy (THz-TDS). THz-TDS is conducted to obtain the frequency-resolved photoconductivity spectra. Instead of only recording the change of the THz peak electric field after the photoexcitation, one can detect the whole THz waveform with ($E_{\text{pump}}(t)$) and without ($E_0(t)$) optical excitations (and thus their difference $\Delta E(t) = E_{\text{pump}}(t) - E_0(t)$) at a fixed pump–sampling delay time by moving the optical delay lines for the pump beam and sampling beam simultaneously.³⁸ By Fourier transformation of $E_0(t)$ and $\Delta E(t)$, the complex photoconductivity spectra $\sigma(\omega)$ are achieved as follows:

$$\sigma(\omega) = \frac{n_1 + n_2}{Z_0 l} \left[-\frac{\Delta E(\omega)}{E_0(\omega)} \right]$$

■ ASSOCIATED CONTENT

Data Availability Statement

All data needed to evaluate the conclusions in the paper are present in the paper and/or the Supporting Information.

SI Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acseenergylett.2c01949>.

UV–vis absorption, determination of photon-to-carrier quantum yield, lifetime estimation at low fluence, discussion on the fast decay of OPTP dynamics, extraction of critical density by THz-TDS, extracted critical densities at different pump wavelengths and

different temperatures, Mott polaron density calculations, photon-to-carrier conversion ratio at different temperatures, and density-dependent effective mass and charge mobility in γ -CsPbI₃ (PDF)

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Author Contributions

H.I.W., J.H., and M.B. conceived and designed the project. H.Z. conducted THz measurements and analyzed the data. E.D. and D.V. synthesized the perovskite samples. B.V.-B., J.M.F., and A.W. conducted the theoretical calculations. H.Z., H.I.W., and M.B. drafted the paper with input from all authors; all authors discussed the results and have given approval to the final version of the manuscript.

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Notes

The authors declare no competing financial interest.

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REFERENCES

- (1) Liu, X. K.; Xu, W.; Bai, S.; Jin, Y.; Wang, J.; Friend, R. H.; Gao, F. Metal halide perovskites for light-emitting diodes. *Nat. Mater.* **2021**, *20*, 10–21.
- (2) Lin, K.; Xing, J.; Quan, L. N.; de Arquer, F. P. G.; Gong, X.; Lu, J.; Xie, L.; Zhao, W.; Zhang, D.; Yan, C.; Li, W.; Liu, X.; Lu, Y.; Kirman, J.; Sargent, E. H.; Xiong, Q.; Wei, Z. Perovskite light-emitting diodes with external quantum efficiency exceeding 20%. *Nature* **2018**, *562*, 245–248.
- (3) Min, H.; Lee, D. Y.; Kim, J.; Kim, G.; Lee, K. S.; Kim, J.; Paik, M. J.; Kim, Y. K.; Kim, K. S.; Kim, M. G.; Shin, T. J.; Il Seok, S. Perovskite solar cells with atomically coherent interlayers on SnO₂ electrodes. *Nature* **2021**, *598*, 444–450.
- (4) Jeong, J.; Kim, M.; Seo, J.; Lu, H.; Ahlawat, P.; Mishra, A.; Yang, Y.; Hope, M. A.; Eickemeyer, F. T.; Kim, M.; Yoon, Y. J.; Choi, I. W.; Darwich, B. P.; Choi, S. J.; Jo, Y.; Lee, J. H.; Walker, B.; Zakeeruddin, S. M.; Emsley, L.; Rothlisberger, U.; Hagfeldt, A.; Kim, D. S.; Gratzel, M.; Kim, J. Y. Pseudo-halide anion engineering for α -FAPbI₃ perovskite solar cells. *Nature* **2021**, *592*, 381–385.
- (5) Zhu, X. Y.; Podzorov, V. Charge carriers in hybrid organic-inorganic lead halide perovskites might be protected as large polarons. *J. Phys. Chem. Lett.* **2015**, *6*, 4758–4761.
- (6) Chen, Y.; Yi, H. T.; Wu, X.; Haroldson, R.; Gartstein, Y. N.; Rodionov, Y. I.; Tikhonov, K. S.; Zakhidov, A.; Zhu, X. Y.; Podzorov, V. Extended carrier lifetimes and diffusion in hybrid perovskites revealed by Hall effect and photoconductivity measurements. *Nat. Commun.* **2016**, *7*, 12253.
- (7) Herz, L. M. Charge-carrier mobilities in metal halide perovskites: Fundamental mechanisms and limits. *ACS Energy Lett.* **2017**, *2*, 1539–1548.
- (8) Kang, J.; Wang, L. W. High defect tolerance in lead halide perovskite CsPbBr₃. *J. Phys. Chem. Lett.* **2017**, *8*, 489–493.
- (9) Karakus, M.; Jensen, S. A.; D'Angelo, F.; Turchinovich, D.; Bonn, M.; Canovas, E. Phonon-electron scattering limits free charge mobility in methylammonium lead iodide perovskites. *J. Phys. Chem. Lett.* **2015**, *6*, 4991–4996.
- (10) Zhang, H.; Debroye, E.; Steele, J. A.; Roeffaers, M. B. J.; Hofkens, J.; Wang, H. I.; Bonn, M. Highly mobile large polarons in black phase CsPbI₃. *ACS Energy Lett.* **2021**, *6*, 568–573.
- (11) Zhu, H.; Miyata, K.; Fu, Y.; Wang, J.; Joshi, P. P.; Niesner, D.; Williams, K. W.; Jin, S.; Zhu, X.-Y. Screening in crystalline liquids protects energetic carriers in hybrid perovskites. *Science* **2016**, *353*, 1409–1413.
- (12) Brenner, P.; Bar-On, O.; Jakoby, M.; Allegro, I.; Richards, B. S.; Paetzold, U. W.; Howard, I. A.; Scheuer, J.; Lemmer, U. Continuous wave amplified spontaneous emission in phase-stable lead halide perovskites. *Nat. Commun.* **2019**, *10*, 988.
- (13) Wang, Z.; Lin, Q.; Wenger, B.; Christoforo, M. G.; Lin, Y.-H.; Klug, M. T.; Johnston, M. B.; Herz, L. M.; Snaith, H. J. High irradiance performance of metal halide perovskites for concentrator photovoltaics. *Nat. Energy* **2018**, *3*, 855–861.
- (14) Shen, J.-X.; Zhang, X.; Das, S.; Kioupakis, E.; Van de Walle, C. G. Unexpectedly strong auger recombination in halide perovskites. *Adv. Energy Mater.* **2018**, *8*, 1801027.
- (15) Jiang, Y.; Cui, M.; Li, S.; Sun, C.; Huang, Y.; Wei, J.; Zhang, L.; Lv, M.; Qin, C.; Liu, Y.; Yuan, M. Reducing the impact of auger recombination in quasi-2d perovskite light-emitting diodes. *Nat. Commun.* **2021**, *12*, 336.
- (16) Yang, Y.; Ostrowski, D. P.; France, R. M.; Zhu, K.; van de Lagemaat, J.; Luther, J. M.; Beard, M. C. Observation of a hot-phonon bottleneck in lead-iodide perovskites. *Nat. Photonics* **2016**, *10*, 53–59.
- (17) Frost, J. M.; Whalley, L. D.; Walsh, A. Slow cooling of hot polarons in halide perovskite solar cells. *ACS Energy Lett.* **2017**, *2*, 2647–2652.
- (18) Joshi, P. P.; Maehrlein, S. F.; Zhu, X. Dynamic screening and slow cooling of hot carriers in lead halide perovskites. *Adv. Mater.* **2019**, *31*, e1803054.
- (19) Edwards, P. P.; Sienko, M. J. Universality aspects of the metal-nonmetal transition in condensed media. *Phys. Rev. B* **1978**, *17*, 2575–2581.
- (20) Bretschneider, S. A.; Ivanov, I.; Wang, H. I.; Miyata, K.; Zhu, X.; Bonn, M. Quantifying polaron formation and charge carrier cooling in lead-iodide perovskites. *Adv. Mater.* **2018**, *30*, e1707312.
- (21) Wang, H. I.; Infante, I.; Brinck, S. T.; Canovas, E.; Bonn, M. Efficient hot electron transfer in quantum dot-sensitized mesoporous oxides at room temperature. *Nano Lett.* **2018**, *18*, 5111–5115.
- (22) Ulbricht, R.; Hendry, E.; Shan, J.; Heinz, T. F.; Bonn, M. Carrier dynamics in semiconductors studied with time-resolved terahertz spectroscopy. *Rev. Mod. Phys.* **2011**, *83*, 543–586.
- (23) Kang, Y.; Han, S. Intrinsic carrier mobility of cesium lead halide perovskites. *Phys. Rev. Appl.* **2018**, *10*, 044013.
- (24) Miyata, K.; Meggiolaro, D.; Trinh, M. T.; Joshi, P. P.; Mosconi, E.; Jones, S. C.; De Angelis, F.; Zhu, X.-Y. Large polarons in lead halide perovskites. *Sci. Adv.* **2017**, *3*, e1701217.
- (25) Sendner, M.; Nayak, P. K.; Egger, D. A.; Beck, S.; Müller, C.; Epding, B.; Kowalsky, W.; Kronik, L.; Snaith, H. J.; Pucci, A.; Lovrinčić, R. Optical phonons in methylammonium lead halide perovskites and implications for charge transport. *Mater. Horiz.* **2016**, *3*, 613–620.
- (26) Frost, J. M. Calculating polaron mobility in halide perovskites. *Phys. Rev. B* **2017**, *96*, 195202.
- (27) Schultz, T. D. Slow electrons in polar crystals: Self-energy, mass, and mobility. *Phys. Rev.* **1959**, *116*, 526–543.
- (28) Ciuchi, S.; Lorenzana, J.; Pierleoni, C. Induced charge in a frohlich polaron: Sum rule and spatial extent. *Phys. Rev. B* **2000**, *62*, 4426–4430.
- (29) Steele, J. A.; Jin, H.; Dovgaliuk, I.; Berger, R. F.; Braeckvelt, T.; Yuan, H.; Martin, C.; Solano, E.; Lejaeghere, K.; Rogge, S. M. J.; Notebaert, C.; Vandezande, W.; Janssen, K. P. F.; Goderis, B.; Debroye, E.; Wang, Y. K.; Dong, Y.; Ma, D.; Saidaminov, M.; Tan, H.; Lu, Z.; Dyadkin, V.; Chernyshov, D.; Van Speybroeck, V.; Sargent, E. H.; Hofkens, J.; Roeffaers, M. B. J. Thermal nonequilibrium of strained black CsPbI₃ thin films. *Science* **2019**, *365*, 679–684.
- (30) Emin, D. Barrier to recombination of oppositely charged large polarons. *J. Appl. Phys.* **2018**, *123*, 055105.
- (31) Miyata, K.; Zhu, X.-Y. Ferroelectric large polarons. *Nat. Mater.* **2018**, *17*, 379–381.
- (32) Milot, R. L.; Eperon, G. E.; Snaith, H. J.; Johnston, M. B.; Herz, L. M. Temperature-dependent charge-carrier dynamics in CH₃NH₃PbI₃ perovskite thin films. *Adv. Funct. Mater.* **2015**, *25*, 6218–6227.
- (33) Yang, Y.; Yang, M.; Li, Z.; Crisp, R.; Zhu, K.; Beard, M. C. Comparison of recombination dynamics in CH₃NH₃PbBr₃ and CH₃NH₃PbI₃ perovskite films: Influence of exciton binding energy. *J. Phys. Chem. Lett.* **2015**, *6*, 4688–4692.
- (34) Fu, J.; Xu, Q.; Han, G.; Wu, B.; Huan, C. H. A.; Leek, M. L.; Sum, T. C. Hot carrier cooling mechanisms in halide perovskites. *Nat. Commun.* **2017**, *8*, 1300.
- (35) La-o-vorakiat, C.; Salim, T.; Kadro, J.; Khuc, M.-T.; Haselberger, R.; Cheng, L.; Xia, H.; Gurzadyan, G. G.; Su, H.;

Lam, Y. M.; Marcus, R. A.; Michel-Beyerle, M.-E.; Chia, E. E. M. Elucidating the role of disorder and free-carrier recombination kinetics in $\text{CH}_3\text{NH}_3\text{PbI}_3$ perovskite films. *Nat. Commun.* **2015**, *6*, 7903.

(36) Emin, D. *Polarons*; Cambridge University Press, 2013.

(37) Kuzel, P.; Kadlec, F.; Nemeč, H. Propagation of terahertz pulses in photoexcited media: Analytical theory for layered systems. *J. Chem. Phys.* **2007**, *127*, 024506.

(38) Zhang, H.; Debroye, E.; Zheng, W.; Fu, S.; Virgilio, L. D.; Kumar, P.; Bonn, M.; Wang, H. I. Highly mobile hot holes in $\text{Cs}_2\text{AgBiBr}_6$ double perovskite. *Sci. Adv.* **2021**, *7*, eabj9066.

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